Spin-Dependent Transport in van der Waals Magnetic Tunnel Junctions with Fe₃GeTe₂ Electrodes

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Supporting Information

ABSTRACT: van der Waals (vdW) heterostructures, stacking different two-dimensional materials, have opened up unprecedented opportunities to explore new physics and device concepts. Especially interesting are recently discovered two-dimensional magnetic vdW materials, providing new paradigms for spintronic applications. Here, using density functional theory (DFT) calculations, we investigate the spindependent electronic transport across vdW magnetic tunnel junctions (MTJs) composed of Fe₃GeTe₂ ferromagnetic electrodes and a graphene or hexagonal boron nitride (h-BN) spacer layer. For both types of junctions, we find that the junction resistance changes by thousands of percent when the



magnetization of the electrodes is switched from parallel to antiparallel. Such a giant tunneling magnetoresistance (TMR) effect is driven by dissimilar electronic structure of the two spin-conducting channels in Fe₃GeTe₂, resulting in a mismatch between the incoming and outgoing Bloch states in the electrodes and thus suppressed transmission for an antiparallel-aligned MTJ. The vdW bonding between electrodes and a spacer layer makes this result virtually independent of the type of the spacer layer, making the predicted giant TMR effect robust with respect to strain, interface distance, and other parameters, which may vary in the experiment. We hope that our results will further stimulate experimental studies of vdW MTJs and pave the way for their applications in spintronics.

KEYWORDS: van der Waals materials, magnetic tunnel junctions, Fe₃GeTe₂, spin-dependent transport, magnetoresistance

he emergence of magnetic two-dimensional (2D) van der Waals (vdW) materials offers exciting opportunities for exploring new physical phenomena and potential applications.^{1,2} Among these materials are Fe_3GeTe_2 ^{3,4} $Cr_2Ge_2Te_6$ ⁵ VSe_{2} ⁶ and CrI_{3} ⁷ where a long-range 2D magnetic order has recently been discovered. Heterostructures based on these vdW materials have revealed interesting functional properties.^{8,9} In particular, 2D magnetic layers have been employed as electrodes or barriers in vdW magnetic tunnel junctions (MTJs), expanding the field of spin-dependent transport in MTJs beyond conventional transition metal ferromagnetic electrodes and oxide barrier layers. A large tunneling magnetoresistance (TMR) in the vdW MTJs is promising for applications in spintronic devices such as hard disk read heads and magnetic random access memories. For instance, using CrI₃ as a barrier layer in graphite/CrI₃/graphite tunnel junctions has been reported to produce a huge TMR over thousands of percent at low temperature.¹⁰⁻¹² In these tunnel junctions, CrI₃ served as a spin-filter and TMR was associated

with a change of the CrI₃ magnetic ordering from antiferromagnetic to ferromagnetic under the influence of an applied magnetic field.¹³ This is different from the "conventional" TMR effect in MTJs where two ferromagnetic electrodes are realigned by an applied magnetic field, resulting in a change of tunneling resistance,¹⁴ and the related effects in ferroelectric tunnel junctions with ferromagnetic electrodes.^{15–17} Very recently, TMR of 160% has been experimentally observed in Fe₃GeTe₂*lh*-BNlFe₃GeTe₂ MTJs at low temperature.¹⁸ The observed effect resembles the conventional TMR associated with the change of magnetic ordering of ferromagnetic electrodes. However, essentially the spindependent transport mechanism in these vdW MTJs fully made of 2D materials has not been elucidated. There are a number of fundamental questions that need to be addressed in

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Figure 1. (a) Side and top views of the atomic lattice of bulk Fe_3GeTe_2 . The dashed rectangular and rhombic shapes denote the crystal unit cell. (b) Majority-spin (black curves) and minority-spin (red curves) band structure of bulk Fe_3GeTe_2 . (c) Spin-resolved density of states of bulk Fe_3GeTe_2 .

order to guide further developments in the field of vdW MTJs. Among these questions are the effect of the spin-dependent electronic structure of Fe_3GeTe_2 on TMR, the importance of the symmetry selection rules that are known to control TMR in MTJs with transition metal electrodes and crystalline tunnel barriers, ^{19–23} the role of the tunnel barrier layer and its electronic structure, ^{19,23} and the effect of the interface bonding on magnetoresistive properties.¹⁸

Answering these questions are especially important from the practical perspective. Among known 2D ferromagnetic vdW materials, Fe₃GeTe₂ exhibits relatively high Curie temperature around 220 K in its bulk state. 4,24-26 Moreover, it has been shown that the Curie temperature of Fe₃GeTe₂ can be raised up to room temperature by ionic gating.⁴ The metallic nature of Fe₃GeTe₂ enables using this material as a magnetic electrode in vdW MTJs, which has advantages over insulating CrI₃ used as a spin-filter barrier. First, in CrI₃-based MTJs, a large magnetic field (~ 1 T) is required to switch the antiferromagnetic ground state to ferromagnetic.^{10–12} Second, CrI₃-based MTJs are volatile, i.e., the magnetic field needs to be maintained to preserve the ferromagnetic order, while Fe₃GeTe₂-based MTJs are nonvolatile due to two stable magnetization configurations (parallel and antiparallel) in the absence of applied field.

Driven by these motivations, in this Letter, we investigate the spin-dependent transport across vdW MTJs with Fe_3GeTe_2 electrodes and two representative nonmagnetic 2D materials graphene (Gr) and *h*-BN as spacer layers. We demonstrate the emergence of a giant TMR in these junctions, which is controlled by the electronic structure of Fe_3GeTe_2 and largely independent of the nature of the spacer layer. These results open interesting perspectives for further experimental exploration of MTJs based on 2D magnetic vdW materials.

First-principles calculations are performed using the Quantum ESPRESSO package²⁷ with PBE-GGA exchange correlation potential²⁸ and ultrasoft pseudopotential.²⁹ The electronic structure of bulk Fe₃GeTe₂ is self-consistently calculated with the lattice parameters being fixed at their experimental values of a = b = 3.991 Å and c = 16.33 Å.²⁴ The Monkhorst *k*-point mesh for the self-consistent calculation is $16 \times 16 \times 4$, and the plane-wave cutoff is 40 Ry.

In a MTJ, the $\sqrt{3} \times \sqrt{3}$ in-plane unit cell of graphene and *h*-BN is matched at the interface with that of the Fe₃GeTe₂ electrode. The interface spacing between Fe₃GeTe₂ and graphene or *h*-BN is relaxed in the presence of the vdW interaction. The electronic structure of Fe₃GeTe₂|GrlFe₃GeTe₂ and Fe₃GeTe₂|*h*-BNlFe₃GeTe₂ supercells is self-consistently calculated using a fine k-point mesh of $16 \times 16 \times 1$. Then, the electron transmission is obtained using the wave function scattering method,³⁰ by matching the wave functions between left and right Fe₃GeTe₂ electrodes and the scattering region, representing MTJs. The two-dimensional Brillouin zone (2DBZ) is sampled using a uniform mesh of 200 × 200 kpoints. The spin-dependent ballistic conductance of the MTJs is obtained by the summation of transmission over the 2DBZ:

$$G_{\sigma} = \frac{e^2}{h} \sum_{\mathbf{k}_{\parallel}} T_{\sigma}(\mathbf{k}_{\parallel})$$

where $T_{\sigma}(\mathbf{k}_{\parallel})$ is the spin and k-resolved transmission probability for an electron at the Fermi energy with spin σ and Bloch wave vector $\mathbf{k}_{\parallel} = (k_x, k_y)$, *e* is the elementary charge, and *h* is the Plank constant. The TMR ratio is defined as TMR = $(G_P - G_{AP})/G_{AP}$, where G_{AP} and G_P are conductances for parallel and antiparallel alignment of magnetization of the Fe₃GeTe₂ electrodes, respectively.

The crystal structure of bulk Fe_3GeTe_2 is depicted in Figure 1a. Fe_3GeTe_2 has a layered hexagonal crystal structure that contains Fe_3Ge slabs separated by the vdW bonded Te layers. The Fe atoms in the unit cell are located in two inequivalent Wyckoff sites, denoted in Figure 1a as Fe1 and Fe2, respectively. The Fe1–Fe1 pair bonds cross a hexagonal network built by covalently bonded Fe2 and Ge atoms. The calculated band structure and density of states of bulk Fe_3GeTe_2 are shown in Figure 1b,c, respectively. The presence of a sizable electron density at the Fermi energy and the exchange splitting of the spin bands indicate that Fe_3GeTe_2 is a ferromagnetic metal. The calculated band structure agrees well with the previous calculation results and the experimental band structures obtained by ARPES.³¹

In a MTJ, transmission and magnetoresistance at zero bias are affected by the spin-polarized Fermi surface of the electrodes. Figure 2a and 2b show the three-dimensional



Figure 2. Majority-spin (a) and minority-spin (b) Fermi surfaces of bulk Fe_3GeTe_2 (plotted using the xcrysden package³²). Different sheets of the Fermi surface, which belong to different bands, are indicated in color. Ballistic transmission for majority-spin (c) and minority-spin (d) channels in bulk Fe_3GeTe_2 , representing the number of conducting channels in 2DBZ indicated in different color.

Fermi surface of bulk Fe_3GeTe_2 for majority- and minorityspin electrons, respectively. It is seen that while the majorityspin Fermi surface (Figure 2a) consists of many Fermi sheets and covers a large portion of the 2DBZ, the minority-spin Fermi surface (Figure 2b) consists of isolated sheets localized in a small portion of the 2DBZ. These bands represent the incoming and outgoing Bloch states, which characterize transport across a MTJ. The number of available Bloch states (conduction channels) at each k_{\parallel} point can be obtained by calculating ballistic transmission in bulk Fe_3GeTe_2 . Figure 2c,d shows the k_{\parallel} -resolved ballistic transmission of bulk Fe_3GeTe_2 for majority- and minority-spin channels, respectively. It is seen that the transmission at each k_{\parallel} point is an integer, mirroring the corresponding spin-resolved Fermi surfaces shown in Figure 2a,b.

By comparing the distribution of the conduction channels in Fe_3GeTe_2 for majority- and minority-spin electrons over the

2DBZ (Figure 2c,d, respectively), we can make qualitative conclusions about TMR in MTJs based on these electrodes, even without knowing the explicit electronic structure of the whole tunnel junction and calculating transmission across it. Due to spin and k_{\parallel} being conserved in the transmission process, transmission between parallel-aligned ferromagnetic electrodes is expected to be much larger than transmission between antiparallel-aligned electrodes. This is due to the fact that in the latter case a mismatch between the incoming and outgoing Bloch states for majority-spin (Figure 2c) and minority-spin (Figure 2d) channels would strongly reduce the transmission. As seen from Figure 2c,d, while the majorityspin channel has multiple bands at the Fermi energy covering the large portion of the 2DBZ, the minority-spin channel has only a few states available, resulting in a large area of the 2DBZ with no overlap. Based on this observation, one can expect high magnetoresistance in 2D vdW MTJs with Fe₃GeTe₂ electrodes.

Next, we explore MTJs based on Fe₃GeTe₂ electrodes and graphene or h-BN barrier layers. The structures of MTJs are shown in Figure S1. Matching of the in-plane lattice leads to graphene and h-BN being compressed by 6.8% and 9.3%, respectively. These large values of the compressive strain are unlikely to occur in the experimental conditions, where the lattices of the electrodes and the barrier would be mismatched due to a weak vdW type bonding, but are required to maintain periodic boundary conditions in our computations. We have compared the band structure of graphene before and after the interface matching and found that there is no significant change in its electronic structure as shown in Figure S2. For h-BN, a slightly larger interface mismatch leads to some changes in the electronic structure, but the insulating nature of this material remains preserved as it is shown in Figure S3. The optimized interface distance between the Fe₃GeTe₂ electrode and the spacer layer is about 3.50 Å, revealing a weak interface bonding across the interface. For comparison, in a FelMgOlFe MTJ, the Fe-MgO interface distance is about 2.16 Å.¹⁹ The weak interface interaction between the electrodes and the spacer layer in vdW MTJs makes the electronic structure of the junctions be simply composed of the electronic structures of its constituents. This behavior is evident from the weightprojected band structure of the Fe3GeTe2|GrlFe3GeTe2 MTJ shown in Figure 3. There are no obvious effects of hybridization and band offset between Fe3GeTe2 and graphene. The graphene bands in the MTJ remain mostly unchanged compared to the isolated graphene. This provides evidence of the rather weak interface interaction between Fe₃GeTe₂ and the spacer layer. We note that while the Dirac cone of graphene lies at the \overline{K} point for the original 1 × 1 cell, it is located at the $\overline{\Gamma}$ point for the $\sqrt{3} \times \sqrt{3}$ unit cell when it is matched to the unit cell of Fe_3GeTe_2 (Figure S2).

Figure 4a–c shows the calculated spin- and k_{II} -resolved electron transmission across a Fe₃GeTe₂[Gr(1 ML)|Fe₃GeTe₂ MTJ for parallel (Figure 4a,b) and antiparallel (Figure 4c) magnetization of the bulk Fe₃GeTe₂ electrodes. It is seen that both majority- and minority-spin transmission resemble the corresponding distribution of the conducting channels in the Fe₃GeTe₂ electrodes (compare Figure 4a to 2c and Figure 4b to 2d). Although the graphene barrier layer filters out electronic transmission at the periphery of the 2DBZ; overall, it does qualitatively change the balance between the majorityand minority-spin contributions. Consistent with our discussion above, the antiparallel transmission (Figure 4c) is



Figure 3. Weight projected band structure in the magnetic junction of $Fe_3GeTe_2|Gr(1 \text{ ML})|Fe_3GeTe_2$. The red circles and solid black circles represent the majority and minority band structures of Fe_3GeTe_2 , respectively. The blue dots demonstrate the projected band weight on graphene. Energy zero indicates the position of Fermi energy.

strongly reduced and qualitatively resembles the minority-spin transmission (Figure 4b).

We note that although graphene is a conductor, transmission across a Fe₃GeTe₂|GrlFe₃GeTe₂ junction is dominated by quantum-mechanical tunneling. The Fermi surface of graphene is reduced to a single $\overline{\Gamma}$ point, and there are no other available propagating states in graphene in the entire 2DBZ. Therefore, for $\mathbf{k}_{\parallel} \neq 0$, the transmission across the graphene layer is controlled by its evanescent states in which decay rate is determined by a barrier height. The latter is increasing away from the $\overline{\Gamma}$ point, resulting in the reduced transmission at the periphery of the 2DBZ.

Figure 4d-f shows the calculated \mathbf{k}_{\parallel} -resolved transmission across a Fe₃GeTe₂lh-BNIFe₃GeTe₂ MTJ. Contrary to graphene, h-BN is an insulator; however, the transmission patterns for the two junctions with graphene and h-BN barrier layers are very similar. Although h-BN does not have propagating states at the Fermi energy, similar to graphene, as it is shown in Figure S3, h-BN has a lower barrier at the center of the 2DBZ ($\overline{\Gamma}$ point), resulting in the enhanced transmission around this area. These results indicate that the key features of spin-dependent tunneling in the vdW MTJs with ferromagnetic Fe_3GeTe_2 electrodes are largely independent of the nature of the barrier. This statement is further confirmed by replacing graphene or *h*-BN with a vacuum barrier layer. The resulting \mathbf{k}_{\parallel} -resolved transmission for a Fe_3GeTe_2 |vacuum|Fe_3GeTe_2 MTJ is shown in Figure S4, which reveals a similar transmission pattern as in the case of Fe_3GeTe_2 |Gr|Fe_3GeTe_ and Fe_3GeTe_1h-BN|Fe_3GeTe_ MTJs.

The total spin-resolved transmission for $Fe_3GeTe_2|Gr|$ Fe_3GeTe_2 and $Fe_3GeTe_2|h$ -BN|Fe_3GeTe_2 MTJs is calculated and summarized in Table 1. The results for MTJ with a

Table 1. Calculated Electron Transmission Across Fe₃GeTe₂|Gr(1 ML)|Fe₃GeTe₂, Fe₃GeTe₂|*h*-BN(1 ML)| Fe₃GeTe₂, and Fe₃GeTe₂|vacuum|Fe₃GeTe₂ MTJs for Parallel (P) and Antiparallel (AP) Magnetization of Fe₃GeTe₂ Electrodes

MTJ structures	P transmission	AP transmission	TMR (%)
Fe ₃ GeTe ₂ Gr Fe ₃ GeTe ₂	2.60×10^{-2}	6.98×10^{-4}	3621
Fe ₃ GeTe ₂ <i>h</i> -BN Fe ₃ GeTe ₂	2.15×10^{-2}	3.38×10^{-4}	6256
$\rm Fe_{3}GeTe_{2} vacuum Fe_{3}GeTe_{2}$	6.18×10^{-3}	4.44×10^{-5}	13814

vacuum barrier are also listed for comparison. It is seen that for both MTJs, the total transmission for the parallel magnetization state is two orders of magnitude larger than that for the antiparallel state. The resulting TMR ratios for MTJs with graphene and h-BN spacer layers are 3621% and 6256%, respectively. These results suggest that in the 2D vdW MTJs, TMR is very large regardless of the spacer layer. This key feature is different from the conventional MgO-based MTJs in which the high TMR ratio originates from the spin-polarized transmission of the Δ_1 -symmetry band across an MgO tunnel barrier.¹⁹ In addition, in traditional MTJs such as FelMgOlFe, the interface effects have crucial importance for the spindependent transport, and the resulting TMR is affected by the interface resonant states.^{20,33} However, in MTJs with 2D ferromagnetic electrodes, the interface effects are expected to be suppressed due to a weak vdW interaction. As a result, the transport properties are mainly determined by the spindependent electronic properties of bulk Fe₃GeTe₂.

Previous studies of MgO-based MTJs have shown that TMR may change sizably depending on the energy position of the



Figure 4. \mathbf{k}_{\parallel} -resolved electron transmission distribution in the 2DBZ for $Fe_3GeTe_2|Gr(1 \text{ ML})|Fe_3GeTe_2 (a-c)$ and $Fe_3GeTe_2|h-BN(1 \text{ ML})|$ $Fe_3GeTe_2 (d-f)$ MTJs. (a,d) Majority-spin transmission for parallel magnetization state; (b.e) minority-spin transmission for parallel magnetization state; (c,f) either-spin transmission for antiparallel magnetization state.

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Figure 5. Electron transmission for the P (black squares) and AP (red circles) magnetization states and the corresponding TMR ratio (blue triangles, refer to the right axis) as a function of energy for $Fe_3GeTe_2|Gr(1 \text{ ML})|Fe_3GeTe_2$ (a) and $Fe_3GeTe_2|h-BN(1 \text{ ML})|Fe_3GeTe_2$ (b) MTJs. The insets show the details of the AP state transmission in the energy range of $E_F - 0.3$ eV to E_F .



Figure 6. Majority-spin (top panel) and minority-spin (bottom panel) Fermi surfaces of Fe_3GeTe_2 at different energies ranging from $E_F - 0.3$ to $E_F + 0.3$ eV, where "0" indicates the Fermi energy. Colors indicate the Fermi surfaces belonging to different bands.

Fermi level. Moreover, in practice, the Fermi level of the junction may be different from the ideal case due to structural imperfections, doping, etc. In order to clarify the effect of the Fermi energy and prove that our conclusions are robust with respect to this characteristic, we calculate the transmission as a function of energy. The transmission and TMR of the Fe₃GeTe₂|Gr|Fe₃GeTe₂ and Fe₃GeTe₂|h-BN|Fe₃GeTe₂ MTJs are calculated within the energy window ranging from -0.3 eV to +0.3 eV (the original Fermi level, $E_{\rm F}$, is set at zero). The calculated transmission and TMR ratio as a function of energy are plotted in Figure 5 and are listed in Tables S1 and S2. We see that the transmission as a function of energy shows similar trends for the two junctions. In both cases, the TMR value is reduced at positive energies but dramatically enhanced at negative energies. In the energy range below $E_{\rm F}$ - 0.2 eV (between $E_{\rm F}$ – 0.2 eV and $E_{\rm F}$ – 0.3 eV in Figure 5), the TMR ratio becomes virtually infinite. This is due to the fact that the antiparallel-state transmission in this energy range becomes zero as is seen in Figure 5. The similar trend is demonstrated by a Fe₃GeTe₂|vacuum|Fe₃GeTe₂ MTJ, where the TMR ratio goes to infinity at negative energies below $E_{\rm F} - 0.2$ eV (Figure **S**5).

We note that the experimentally measured values of TMR at small bias voltage in $Fe_3GeTe_2|h$ -BN| Fe_3GeTe_2 MTJs are around 160% at 4.2 K,¹⁸ which is far lower than the predicted TMR. The shift of the Fermi level in real junctions may be the

reason for the reduced TMR. In addition, the experimentally observed lower values of TMR may result from misaligned Fe_3GeTe_2 electrodes in realistic structures. While the relative translation of the spacer layer and one of the electrodes make a minor effect on the transport properties and TMR, we find that relative rotation of the electrodes with respect to each other leads to the reduction of TMR (Section 7 in Supporting Information). These results indicate that there is much more room for improvement of the magnetoresistive properties of vdW MTJs with Fe_3GeTe_2 electrodes by optimizing the MTJ structures.

A complete mismatch between the Fermi surfaces of the two spin channels is the reason for zero transmission for the antiparallel magnetization. A similar behavior is expected to occur in MTJs with half-metallic ferromagnets, such as Heusler alloys Co₂MnSi, Co₂MnGe, etc., which have only one spin band at the Fermi energy. However, in MTJs with half-metallic Heusler electrodes, e.g., Co₂MnSilMgOlCo₂MnSi, interface states diminish half-metallicity of the Heusler alloy through the interface bonding.³⁴ Thanks to the weak interface interaction in 2D vdW MTJs, the interface bonding effects are suppressed, and the transmission is entirely controlled by the bulk properties of ferromagnetic electrodes. Figure 6 shows the calculated Fermi surface of Fe₃GeTe₂ at different energies. It is seen that in the energy range of $E_{\rm F} - 0.2$ and $E_{\rm F} - 0.3$ eV, the majority- and minority-spin conduction channels do not

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overlap in the 2DBZ. As a result, there are no electrons that can successfully pass through the MTJ when the magnetization of the Fe_3GeTe_2 electrodes is antiparallel, which leads to the infinite TMR in the corresponding energy range.

Using the calculated transmission as a function of energy, it is straightforward to calculate the I-V curves. Under a small bias voltage, ignoring the nonequilibrium effect, the electric current per unit cell under bias voltage V is given by

$$I_{\rm P,AP} = \frac{e}{h} \int_{E_{\rm F}-{\rm eV}/2}^{E_{\rm F}+{\rm eV}/2} T_{\rm P,AP}(E) \, {\rm d}E$$

Here $T_{\rm P}(E)$ $(T_{\rm AP}(E))$ is the transmission as a function of energy E for the P (AP) state of MTJ (Figure 5). The on/off ratio can be defined as the ratio between the currents for the P and AP states: I_P/I_{AP} . Figure S6 shows the calculated I-Vcurves for the two MTJs. It is seen that the I-V curves reveal generally linear bias dependence for the P state of the MTJs and nonlinear bias dependence for the AP state. Such bias dependence behaviors agree with the experimental measurements.¹⁸ In addition, the sharp decrease of TMR with the increasing bias voltage is observed experimentally.¹⁸ The bias voltage at which the TMR drops to half of its value at zero bias is less than 0.05 V. From our theoretical calculations shown in Figure S6, the bias dependence of the on/off ratio demonstrates a similar trend, and the bias voltage at which the on/off ratio drops to half of its value at zero bias is estimated to be 0.04 V.

In summary, the spin-dependent transport properties in 2D vdW magnetic tunnel junctions with Fe₃GeTe₂ electrodes have been investigated using first-principles calculations. It is predicted that, regardless of the spacer layer, Fe₃GeTe₂|Grl Fe₃GeTe₂ and Fe₃GeTe₂*lh*-BNlFe₃GeTe₂ MTJs exhibit a TMR ratio as high as thousands of percent, which is the consequence of the significantly different Fermi surfaces of the two spinconduction channels. The key mechanism responsible for the giant TMR effect in the studied vdW MTJs solely relies on the spin-dependent electronic structure of ferromagnetic Fe₃GeTe₂ electrodes. It is also found that in a certain energy range the complete mismatch of the majority- and minority-spin Fermi surfaces results in zero transmission for the antiparallel magnetization state and therefore leads to the infinite TMR. Our results provide an important insight for further experimental investigations of 2D vdW MTJs, which may lead to useful applications in spintronics.

ASSOCIATED CONTENT

S Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.nano-lett.9b01506.

Magnetic junction structures with graphene, *h*-BN, and vacuum sandwiched between two Fe₃GeTe₂ electrodes; the band structure of $\sqrt{3} \times \sqrt{3}$ graphene before and after it is compressed by 6.8%; the band structure of $\sqrt{3} \times \sqrt{3}$ *h*-BN before and after it is compressed by 9.3%; spin-dependent transport properties in Fe₃GeTe₂| vacuumlFe₃GeTe₂ magnetic junction; transmission and *RA* product data as a function of energy in Fe₃GeTe₂|Grl Fe₃GeTe₂ and Fe₃GeTe₂|*h*-BN|Fe₃GeTe₂ magnetic junctions; *I*-*V* curves in Fe₃GeTe₂|GrlFe₃GeTe₂ and Fe₃GeTe₂ magnetic junctions; discus-

sions on the effect of junction structural imperfections (PDF)

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Notes

The authors declare no competing financial interest.

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